

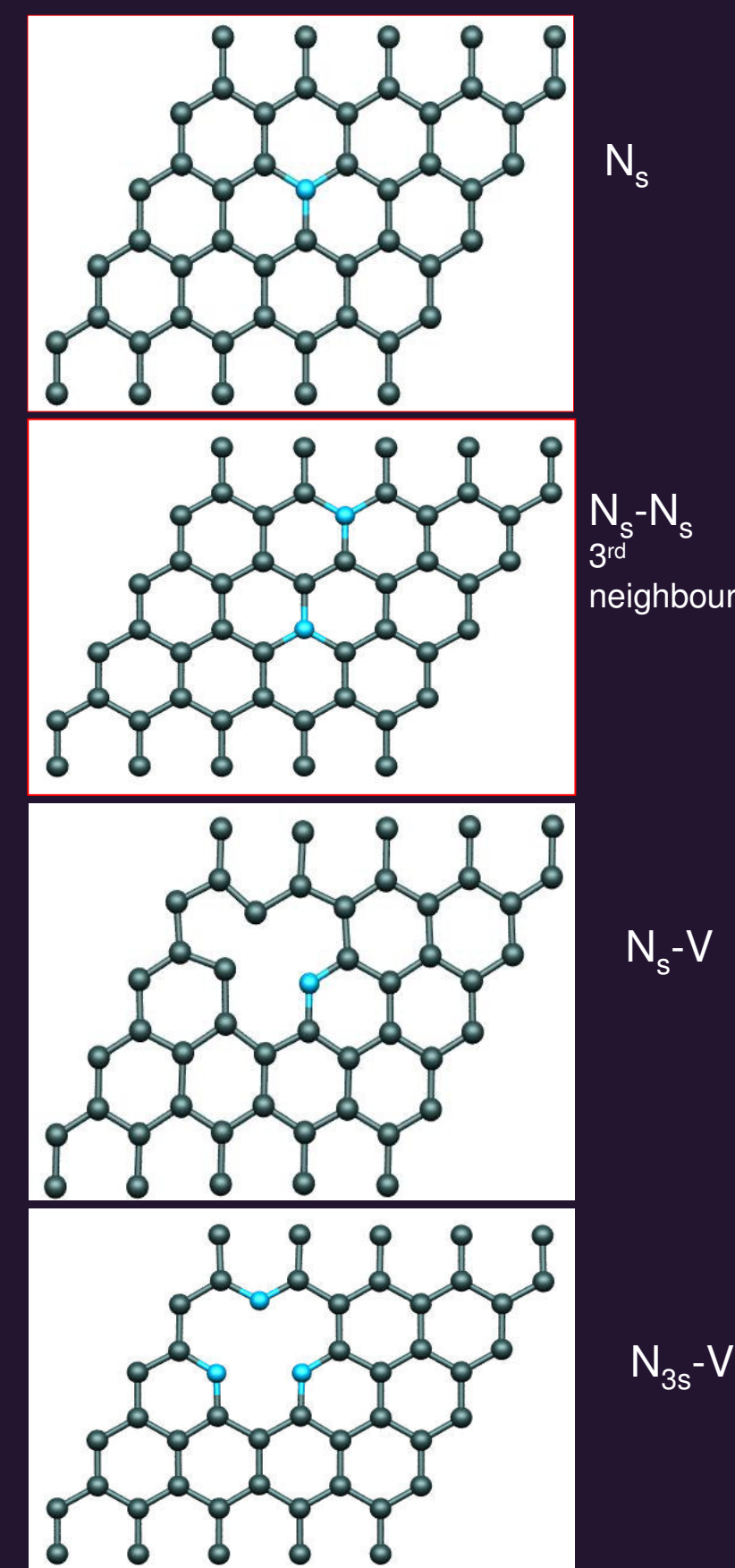
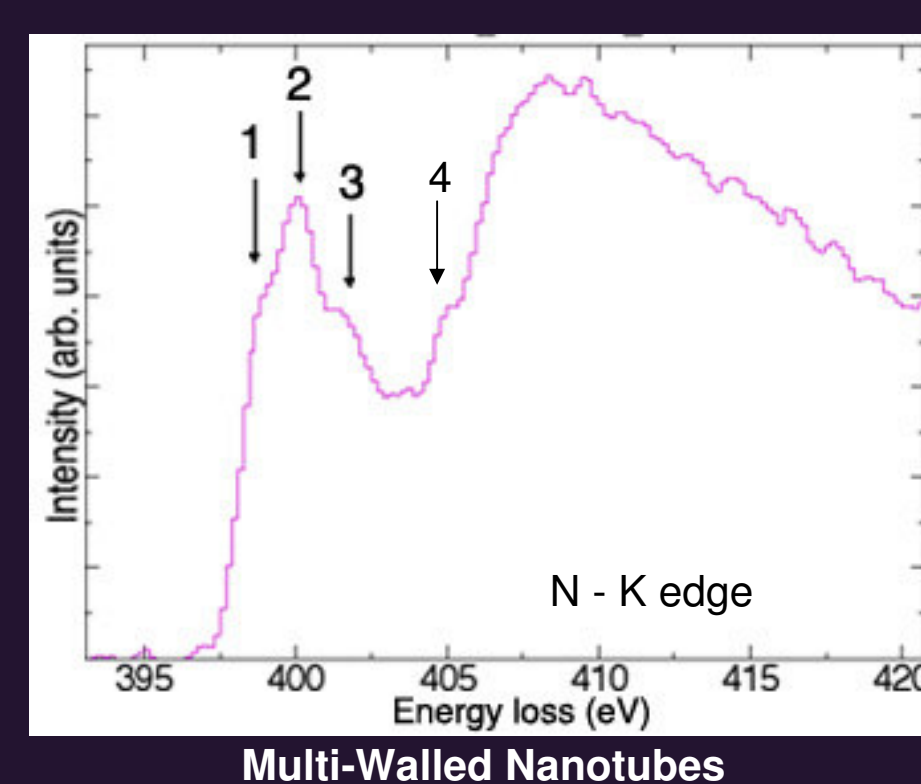
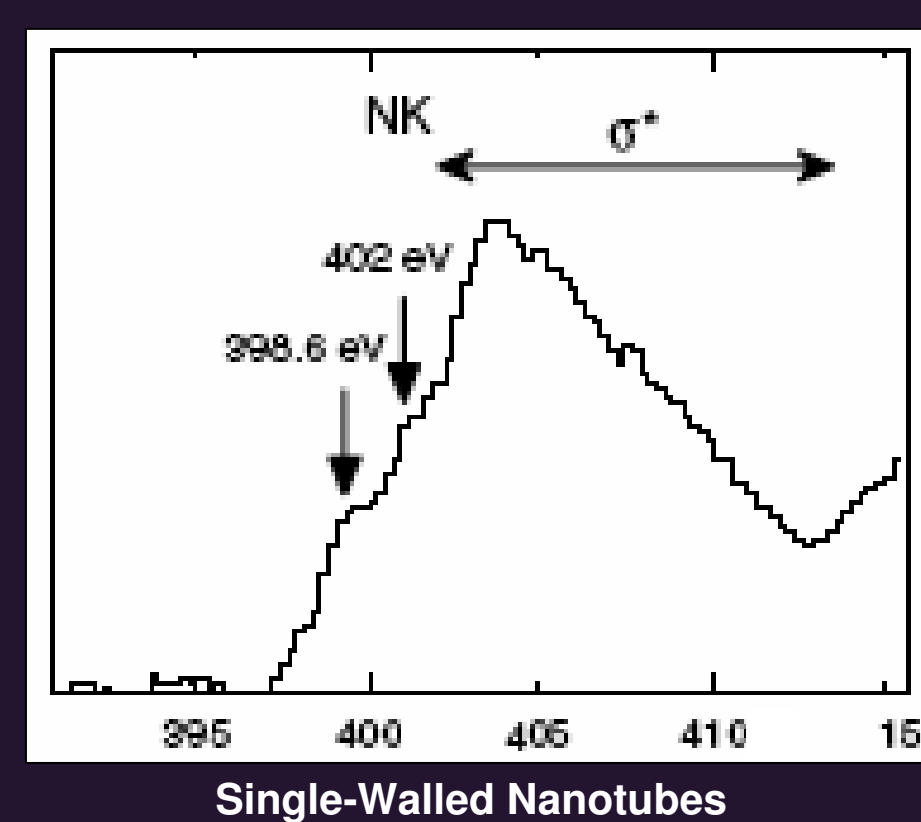
Defects do not just *influence* the behaviour of a material, they often *define* it. Early theoretical nanotube studies focussed on idealised structures, often predicting properties (e.g. Young's Modulus) wildly at odds with experimental results. Only more recently has the *crucial role of defects* begun to be acknowledged, and latterly, exploited.

This work focuses on the study of atomic scale defects and doping in materials, notably in carbon materials such as *graphite, fullerenes and carbon nanotubes*. As well as explaining intrinsic point defect structures, this leads to new techniques such as using irradiation as a strengthening mechanism for nanotubes. This builds on earlier graphite studies showing a new range of *intrinsic point defects* which bridge the gap between atomic layers in graphite. These explain the cause of a major fire at the Windscale nuclear reactor in the 1950s. More recent studies on *doping and functionalisation* of carbon nanotubes show how nitrogen violates one of the tenets of carbon science (the 'isolated pentagon rule'), resulting in a new family of nitrogen doped fullerene molecules, and how *fluorination* can lead to chemical superlattices on nanotube surfaces.

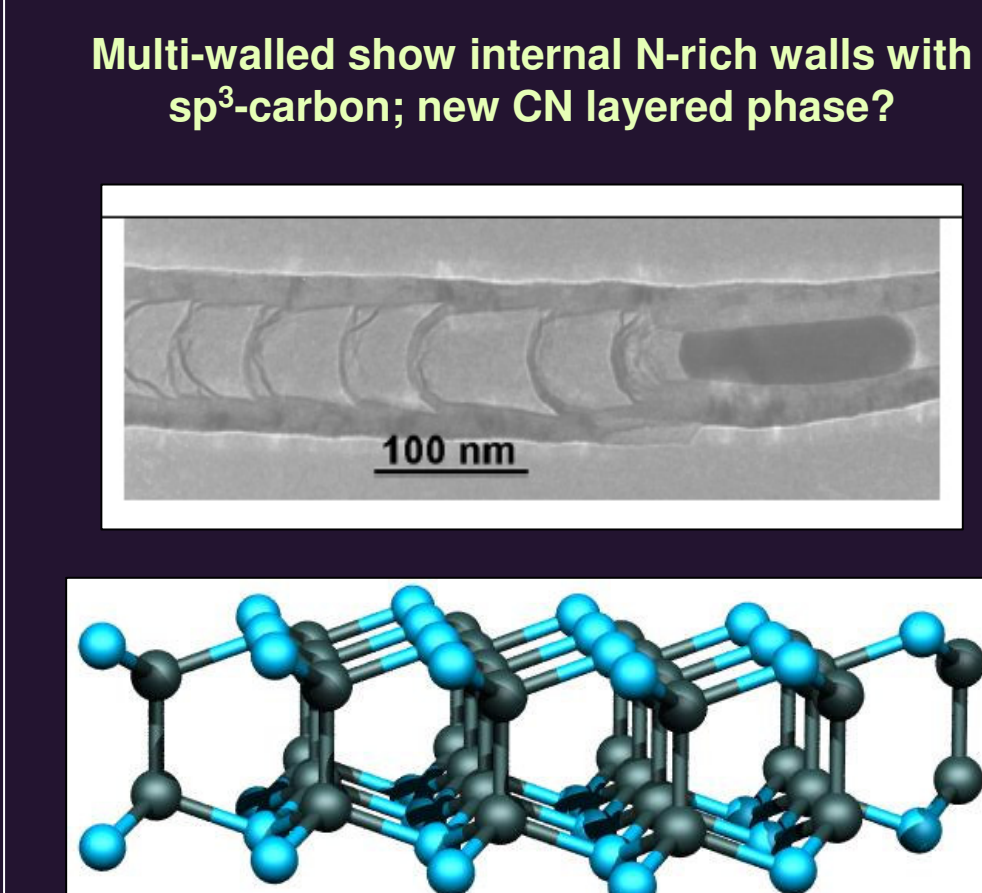
## N and P doping of Carbon Nanotubes : nanoelectronics

EELS of nitrogen doped carbon nanotubes shows

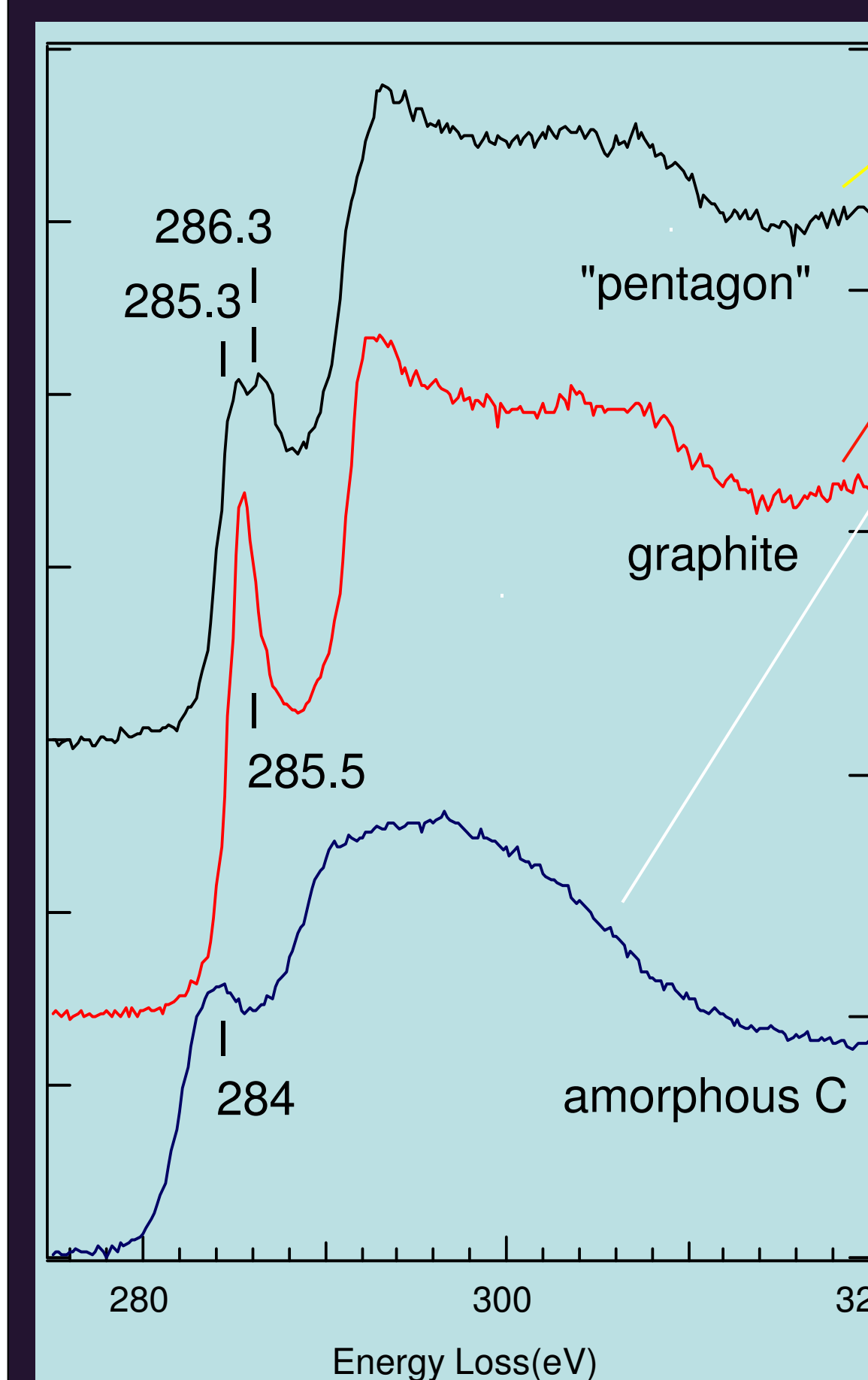
- two N-species in Single-Walled tubes (peaks 1,3)
- a further N-species in Multi-Walled tubes (peaks 2,4)



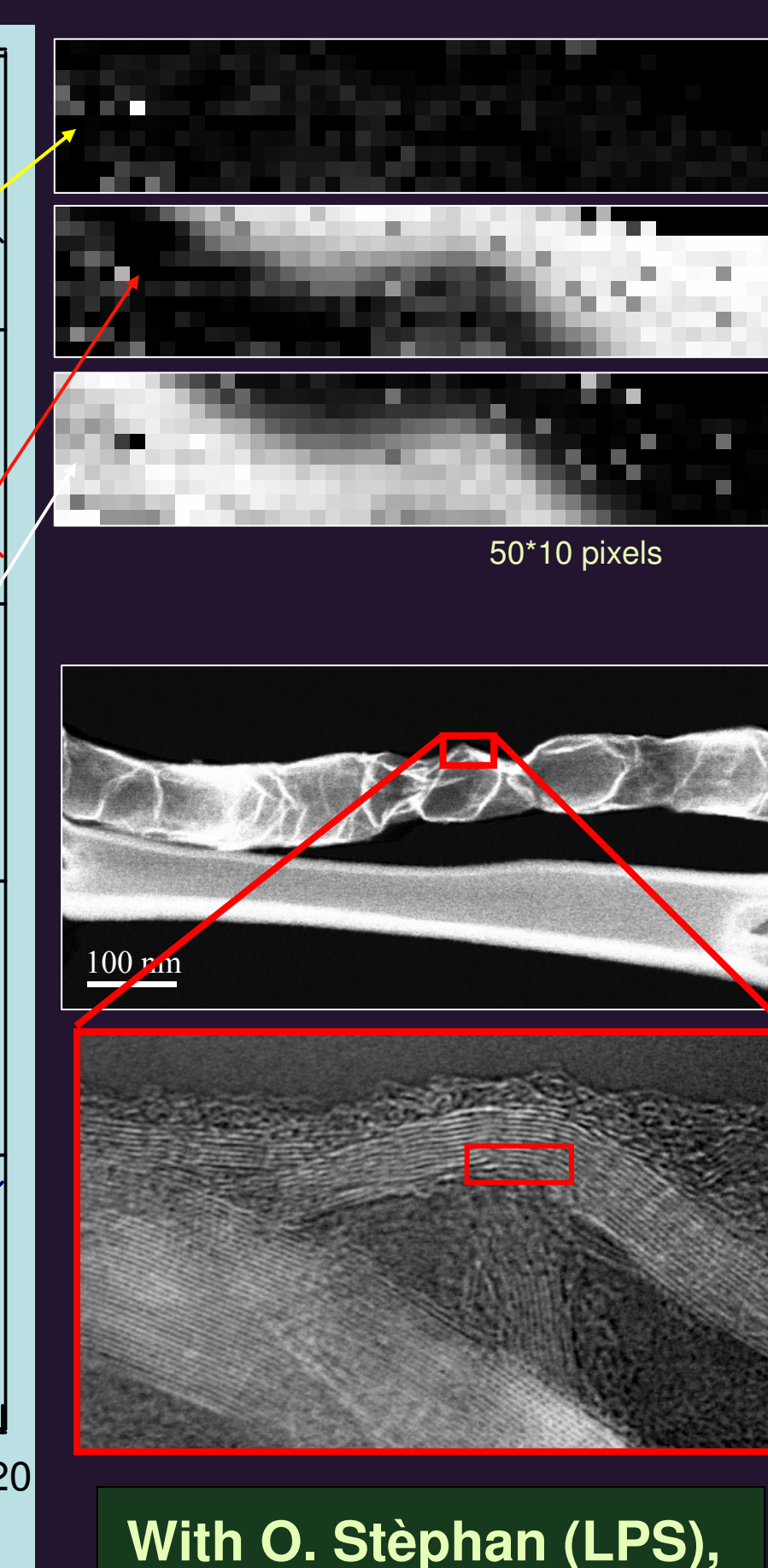
Defect	Formation Energy (eV)
Substitutional N	0.42
N pair at 3rd neighbour	0.67
N - V	5.52
N 2 - V	4.27
N 3 - V	2.53
Graphite Vacancy	8.15
Stone Wales defect	5.63



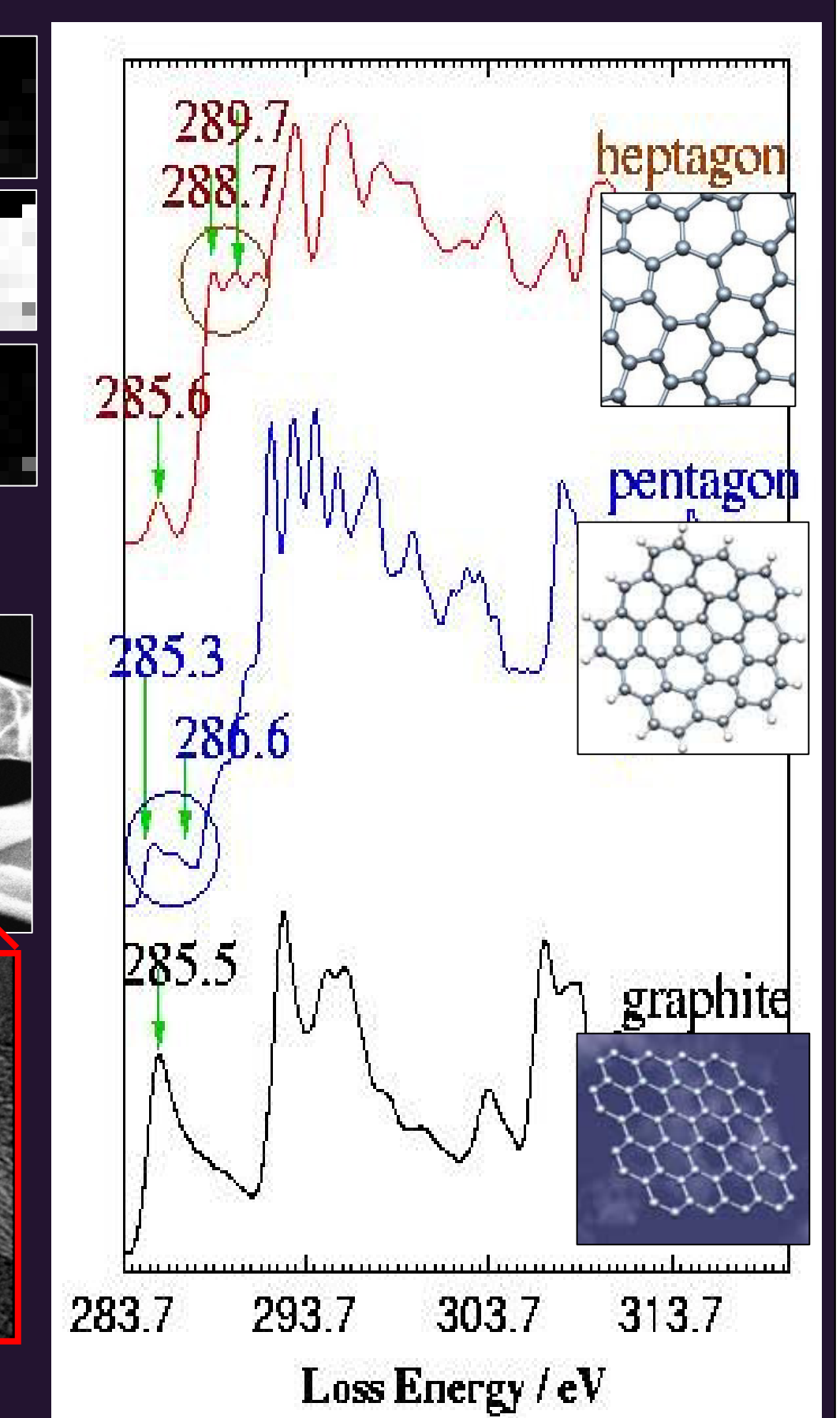
## Can we identify individual defects with EELS?



Experimental EELS



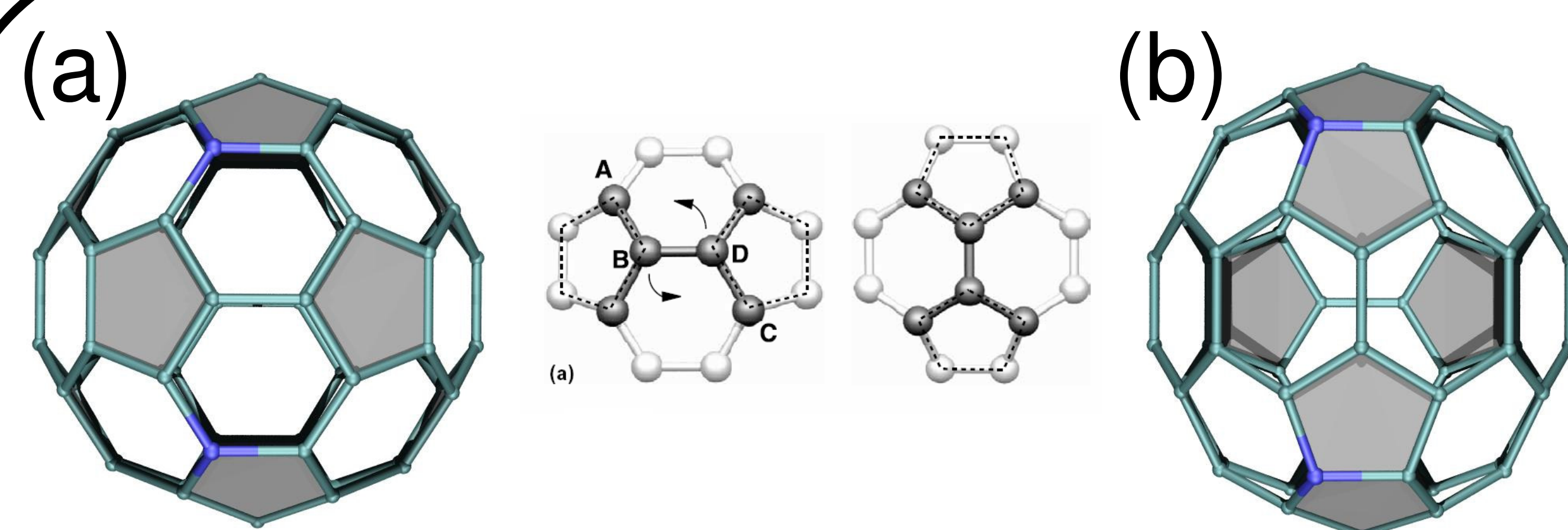
With O. Stéphane (LPS),  
S. Trasobares (Cadiz),  
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M. Heggie (Sussex)



Theoretical EELS

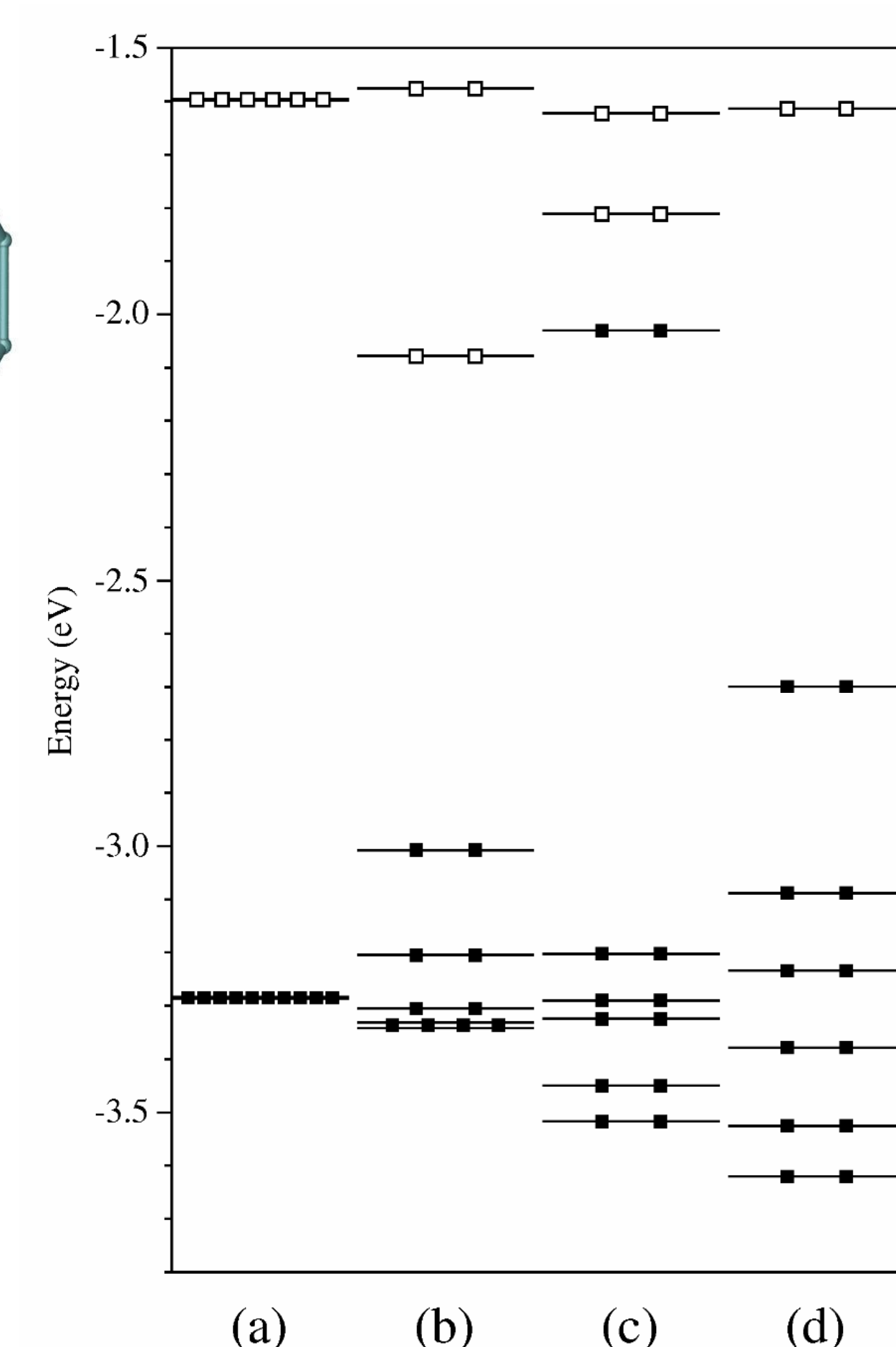


(7,0) carbon nanotube with two nitrogen atoms along the axis (marked in blue), with single vacancy between them. Vacancy formation energy <2.5eV (normally >5eV).



$C_{58}N_2$  (a) before and (b) after the central hexagon bond is rotated by 90°. The right hand structure contains two pentagon pairs, and is **more stable** by 0.54eV. Nitrogen atoms are marked in blue, pentagons are filled in grey. In classic  $C_{60}$  (b) is 1.6eV **less** stable.

Nitrogen stabilises the paired pentagons, and **violates the isolated pentagon rule**



DFT Kohn-Sham eigenvalues (eV) around the HOMO-LUMO gap

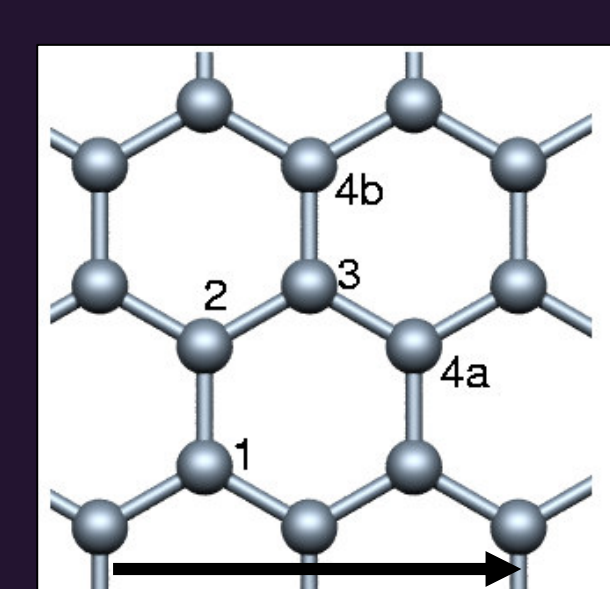
- (a)  $C_{60}$
- (b)  $C_{60}$  with one bond rotation (2 sets of paired pentagons)
- (c) As (a) with two substitutional N atoms
- (d) As (b) with two substitutional N atoms, one per paired pentagon.

Black (white) squares show filled (empty) states.

- In isolated pentagon case, additional N electron forced to occupy high lying anti-bonding state.
- In paired pentagon case, filled level drops. Pentagon pairs become aromatic.

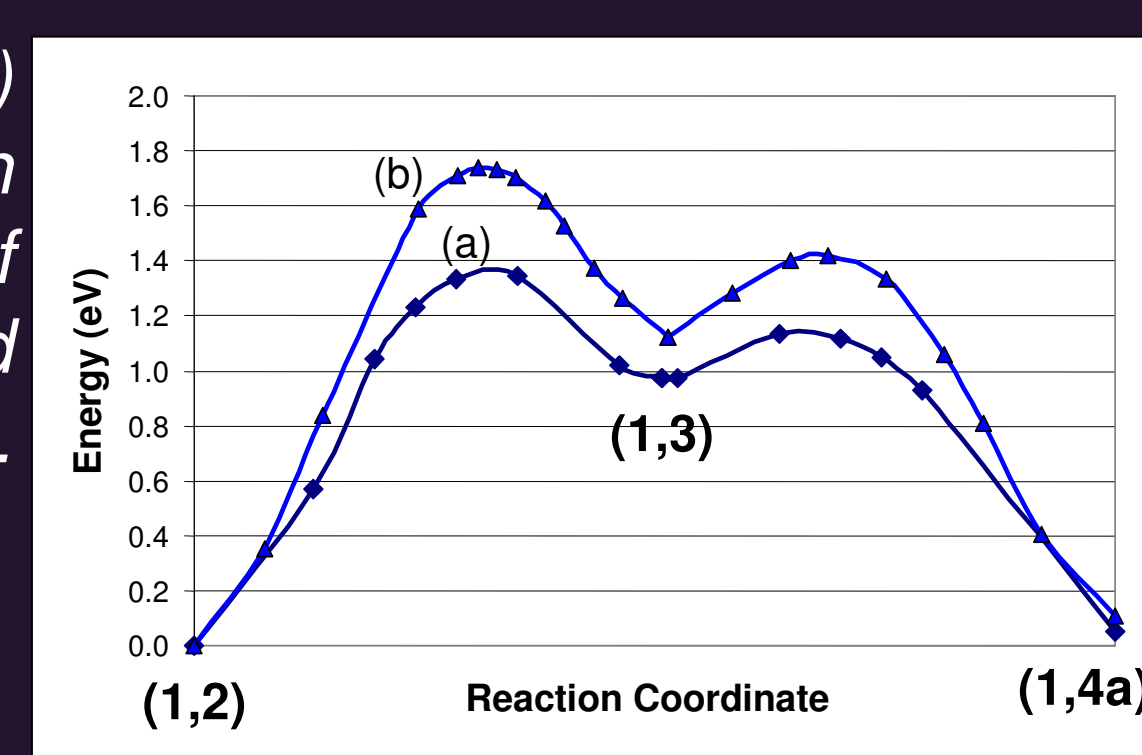
## F<sub>2</sub> addition to an (8,8) nanotube

2 F atoms prefer to sit at (1,4a) and (1,2), strongly covalent (see Figure). (1,3) is much less stable (comparable to isolated F binding). Little distortion of tube – isolated F and F<sub>2</sub> source of semi-ionic XPS F signal.



Migration barriers (eV) for fluorine pair migration on the surface of (a) graphite and (b) (8,8) carbon nanotube.

Arrow marks nanotube axis (x,y) indicates 2 F atoms bonded to carbon atoms x and y

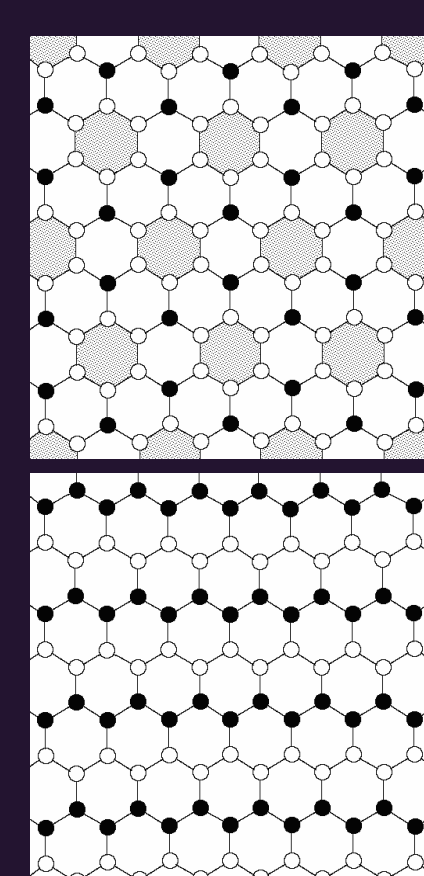


Isolated F migration barrier is low (~0.7eV) Migration barriers for (1,4) → (1,3) → (1,2) for graphite and nanotube are in range corresponding to a 200-250 °C temperature regime.

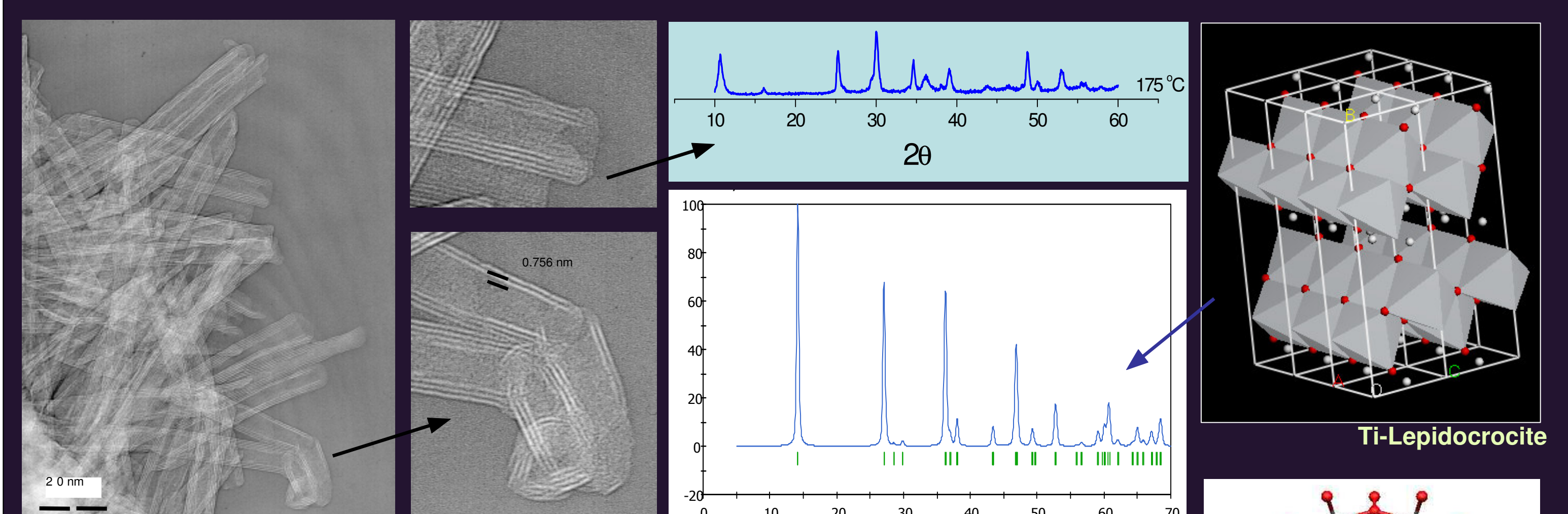
Fluorination leads to different surface superlattices:

Below 200-250 °C restricted migration limits to **C<sub>4</sub>F coverage**

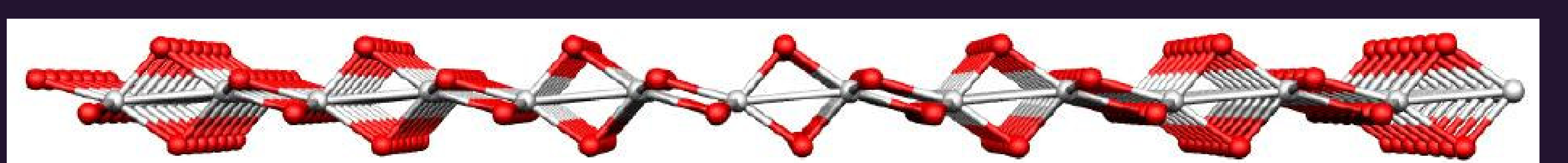
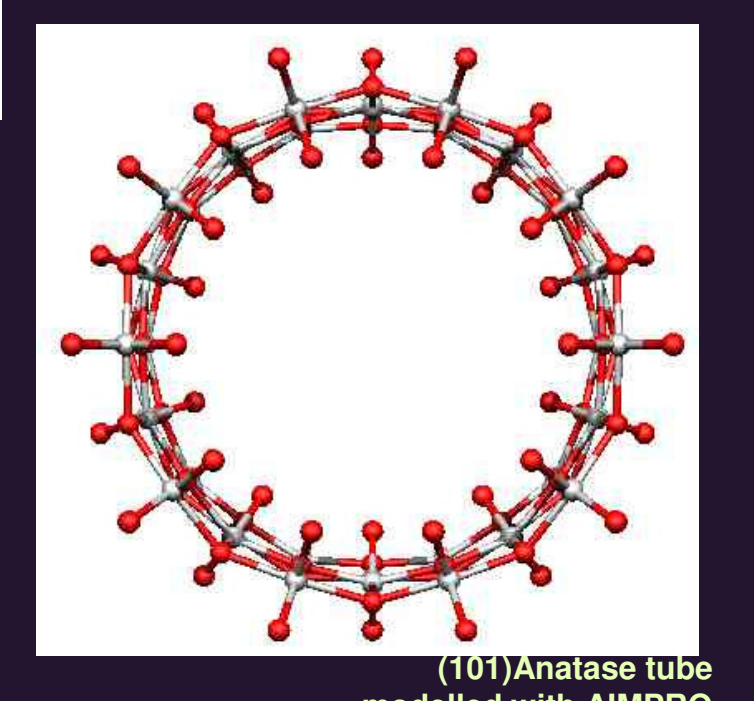
Above 200-250 °C, free motion leads to **C<sub>2</sub>F banded coverage**



## Ti/O nano-scroll structure and surface behaviour



Scroll structure related to (101) anatase Modelling possible structures (TiO<sub>2</sub>, Ti<sub>3</sub>O<sub>7</sub>, TiOOH, H<sub>2</sub>Ti<sub>3</sub>O<sub>7</sub>, lepidocrocite), comparing with XRD, e-diff, TEM. DFT modeling of tubes and gas absorption on surfaces. Material important for photocatalysis, gas sensors, detoxification, H<sub>2</sub> production, etc



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